



Numerical evaluation of temperature fields and residual stresses in butt weld joints and comparison with experimental measurements

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Abstract

This paper presents a novel numerical model, based on the finite element (FE) method, for the simulation of a welding process aimed to make a two-pass V-groove butt joint, paying attention on the prediction of residual stresses and distortions. The 'element birth and death' technique for the simulation of the weld filler supply has been considered within this paper. The main advancements with respect to the state of the art herein proposed concern: (i) the development of a modelling technique able to simulate the plates interaction during the welding operation when an only plate is modelled. This phenomenon is usually neglected in literature; (ii) the heat amount is supplied to the FEs as volumetric generation of the internal energy, allowing overcoming the time-consuming calibration phase needed to use the Goldak's model, commonly adopted in literature. Predicted results showed a good agreement with experimental ones.

KEYWORDS

butt weld joint, element 'birth and death' technique, FEM, residual stress, welding

1 | INTRODUCTION

Welding is among the most relevant joining techniques used in the structural field, and it is particularly attractive for the transport field such as aerospace, automotive and rail, thanks to the advantages it offers in terms of weight saving, monolithic structures, design flexibility and costs. Notwithstanding such benefits, several issues could arise and compromise the efficiency and the performance of the structure. Specifically, defects, residual stresses, porosities, cracks propagation facilitation, distortions and the consequent misalignments of the joint can affect the monoliths due to the thermal cycles involved during the process, as widely described in several books by some authors such as Gunnert¹ or Connor.² The highly

localized transient heat and the strongly nonlinear temperature fields, characterizing the thermal cycles, combined with the subsequent non-uniform cooling phase, cause plastic deformations in both the fusion zone (FZ) and the heat-affected zone (HAZ), as a result of the non-uniform thermal expansion and contraction of the metal.³ Hence, at the end of the welding process, the structure will be characterized by residual stresses that, combined with the in-service loading conditions, could reduce the structural performance, cause assembly issues and influence the fatigue and buckling strength.⁴⁻⁶ Therefore, the measurement of the residual stress-strain state of welded components supports the designers in the development of more efficient structures.

In fact, as many researchers investigated on these issues, there is an extensive literature concerning the evaluation techniques of the residual stresses in welded joints. Wide literature reviews have been proposed by Mackerle⁷ and by Dong.⁸

Over the last years, several destructive and non-destructive techniques have been developed to experimentally evaluate the residual stresses.^{9–16} Among these techniques, the most used ones are the non-destructive ultrasonic techniques, used, for example, by Satyambau and Ramachandran,⁹ and the non-destructive neutron^{12,13} and X-ray¹⁴ diffraction techniques and the destructive hole drilling technique, used by Schajer.¹⁶ However, these techniques show several limits such as the inaccuracies affecting the measures and the high costs. Current computational methods allow overcoming these limitations by simulating the welding processes and determining the stress–strain state; among these, finite element method (FEM) appears to be the most suitable.

During the last decades, several scientific articles proposed finite element (FE) models able to simulate complex welding processes. Typically, due to the coexistence of thermal and mechanical phenomena, the development of numerical models for welding structures can be very challenging; so several strategies could be applied.

A comparison between the modelling strategies based on FEM has been proposed by Mollicone et al¹⁷ in 2006, whereas Lindgren, in 2001, presented a detailed review about the state of the art related to the FE modelling and to the simulation of the welding processes in three articles.^{18–20} Among the many developed techniques, the so-called ‘element birth and death’ is one of the most used. Briefly, it starts by the modelling of the entire weld seam. Subsequently, the FEs of the seam are deactivated and progressively reactivated only when the heat is supplied, as explained in detail in Section 2. The literature presents various researches based on the use of such simulation technique, for different welding processes.

Teng and Chang^{21,22} used the element birth and death technique for simulating the welding process for butt joints made of carbon steel. They used the X-ray diffraction technique for validating the numerical results. Based on the same technique, Armentani et al,^{23–25} in three consecutive studies between 2006 and 2007, simulated the welding processes for butt-welded joints by varying such properties as the weld filler and the thermal material properties. In 2014, the same FE model was established against some experiments.²⁶ Kermanpur et al²⁷ investigated on butt-welded joints, for pipe applications, by using the element birth and death approach. They validated the numerical model against some experimental tests and performed a further sensitivity analysis by changing the arc efficiency and the heat source values

in input. Subsequently, Sepe et al^{28,29} used the same technique for simulating the welding processes of two butt-welded joints, made of similar and dissimilar materials, respectively.

More recently, Mondal et al³⁰ in 2017 investigated on the residual stresses in a multipass welded T-joint by using a FE model, based on the element birth and death technique, validated by comparing numerical and experimental results. Hashemzadeh et al³¹ focused their investigation on butt-welded joints. Finally, Tsirkas,³² in 2018, proposed and validated an element birth and death technique-based model for simulating a laser welding process for aircraft components made of aluminium. With respect to the same technique, Ferro et al^{33,34} investigated numerically on the residual stresses distribution induced by TIG-dressing. Other modelling and simulation techniques have been developed and applied for welding processes. Choi and Mazumder³⁵ proposed a 3D transient FE model for simulating a gas metal arc welding (GMAW) process. Tsirkas et al,³⁶ in 2003, proposed a 3D FE model for simulating a laser welding process, considering also the metallurgical transformations, by using a nonlinear heat transfer analysis, based on a keyhole formation model, and a coupled transient thermomechanical analysis. Similarly, Cho et al³⁷ investigated on the residual stresses in a butt-welded joint and validated the FE model by means of experimental tests based on the hole drilling technique. Gery et al³⁸ carried out a thermal FE simulation of a butt joint developed by a metal inert gas (MIG) welding process and studied the influence of the welding parameters on the temperature fields. Citarella et al^{39,40} developed a dual boundary elements method (DBEM)-based model and a coupled FEM/DBEM for investigating the influence of the residual stresses on the cracks propagation in friction stir welded aluminium butt joint. Other simulation techniques are based on analytical models, as proposed by Mochizuki et al,⁴¹ that evaluate the residual stresses in a pipe butt-welded joint and validate the model by means of a neutron diffraction experimental test. Similarly, another analytical model for friction stir welding has been proposed by Vilaça et al,⁴² whereas Binda et al⁴³ proposed a semi-empirical model, based on analytical solving approach, for simulating a laser welding process and evaluating the temperature fields.

Almost all of the aforementioned simulation models use the Goldak's model^{44,45} to solve the thermal and the mechanical equations, considering either the double-ellipsoid heat source model or the Gaussian heat source model. Nevertheless, Goldak's model requires an extremely accurate calibration phase before proceeding with the simulation of the entire process. This calibration is based on experimental measurements and requires

several control cycles, representing a time-consuming process.⁴⁶

In this study, a novel FE model, based on the ‘element birth and death’ technique, has been developed by means of ABAQUS® v. 6.14 code for the simulation of a welding process that can be applied for several types of joints (e.g., butt joint and T-joint). Among the main proposed elements of novelty, the modelling of the heat input has to be mentioned. The heat amount is supplied to the FEs as volumetric generation of the internal energy. Such technique does not require any calibration phase as for Goldak’s model,^{17,27,32,35,36,46} so the modelling time is significantly reduced.

A two-pass V-groove butt-welded joint, involving two plates characterized by the same material and geometry, has been investigated herein. Taking advantage of the joint symmetry, the FE model has been developed by modelling an only plate and a half seam to reduce the computational costs. Concerning the mechanical analysis, a new modelling strategy is proposed. It consists in simulating the interaction between the two joint counterparts, never considered in the FE models presented in literature.^{17,21,22,31,36,38}

In order to assess the reliability of the proposed numerical procedure, numerical results have been compared with those provided by an experimental test, herein presented. For such purpose, temperatures distribution has been measured during the welding process by using some thermocouples placed at different locations nearby the weld bead; welding distortions have been subsequently measured by means of a coordinate measuring machine (CMM). A very good agreement has been achieved, demonstrating the efficiency of the proposed model.

2 | MATERIALS AND METHODS

Two carbon steel plates of size 248 mm × 125 mm (thickness of 8 mm) which form a single V-groove joint between them (Figure 1A) have been welded by using

the shielded metal arc welding (SMAW) process. The material of the plates is a structural low carbon steel S275JR. The typical chemical composition of the material used in the experimental test and the mechanical properties at room temperature are reported in Tables 1 and 2, respectively. The welding process has been carried out through two passes and a time gap of 108 between the successive passes has been addressed to remove the slag formed during the first pass. The welding parameters, related to each pass, are reported in Table 3. Both weld passes have been carried out at uniform speed and under room conditions using a 3.2 mm diameter flux-coated SMAW electrode ESAB OK 48.50 (AWS E 7018). The weld bead sequence is shown in Figure 1B, and the start point (A) and the end point (B) of each welding pass are shown in Figure 1C. The plates have been simply placed on the work table shown in Figure 1D. In this arrangement, the most parts of the top and bottom surface areas of the plates are exposed to the environmental conditions.

In order to assess the reliability of the proposed FE model and, in particular, the validity of the thermal simulations results, six *K*-type thermocouples have been arranged at different distances (Figure 2A,B) from the weld beads in order to monitor the temperatures distribution: two thermocouples, TC1 and TC2, at a quarter region of the plate and at the mid-plane level ($y = 62$ mm, $z = 4$ mm); the remaining, TC3, TC4, TC5 and TC6, have been fixed at the middle region of the plate and at the mid-plane level ($y = 124$ mm, $z = 4$ mm). To record the values of temperatures, a personal computer with a PCI 6221 DAQ (Data Acquisition) card of National Instrument and LabView® 2018 software have been used. Moreover, during the welding process, both voltage and current have been measured using a voltmeter and an ammeter, respectively, both connected to the weld circuit.

An arc efficiency, η , of 0.82 for SMAW⁴⁷ has been considered; therefore, the heat input per millimetre of weld length, Q , can be calculated using Equation 1:

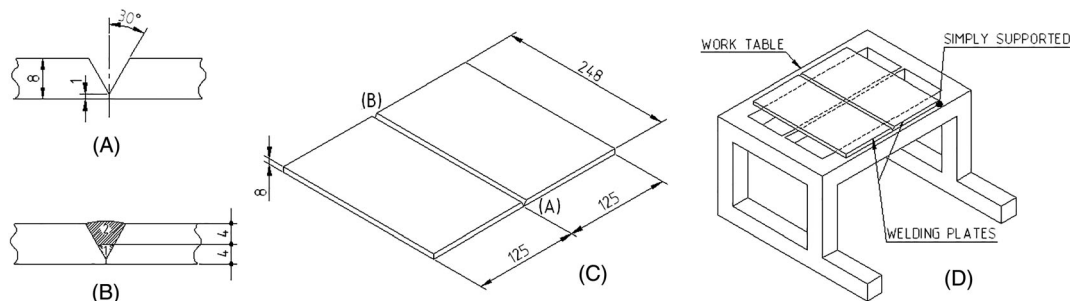


FIGURE 1 Dimensional details of plates and boundary conditions used during the welding process in the experimental test

TABLE 1 Chemical composition (wt%) of S275JR steel

C	Mn	P_{max}	S_{max}	C_{eqmax}
0.12 ÷ 0.18	0.3 ÷ 0.6	0.04	0.05	0.28

TABLE 2 Mechanical properties of the base materials at room temperature

Material	σ_s (MPa)	σ_r (MPa)	E (GPa)	G (GPa)	ν
S275JR	275	430	210	89.8	0.3

TABLE 3 Welding parameters

Pass	Efficiency η	Wire diameter (mm)	Current I (A)	Voltage V (V)	Travel speed v (mm s ⁻¹)	Q (J mm ⁻¹)	\dot{u}''' (W mm ⁻³)
1	0.82	3.2	108	29.5	1.56	1674.7	62.372
2			98	26.0	2.166	964.62	23.829

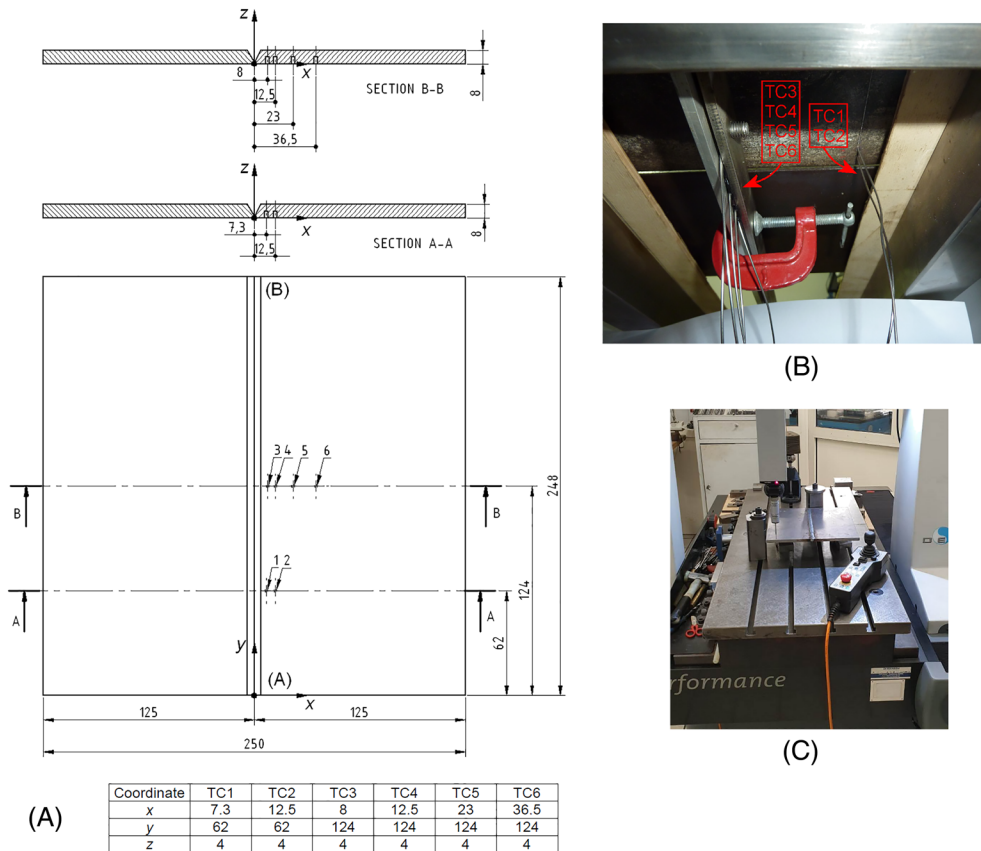


FIGURE 2 (A) Layout of thermocouples, (B) thermocouples installed and (C) distortion of the welding joint measured by means of a coordinate measuring machine (CMM)

$$Q = \eta VI/v, \tag{1}$$

where η is arc efficiency, V is the voltage, I is the current and v is the welding speed. The values of Q are also reported in Table 3.

In order to assess the reliability of the proposed FE model to evaluate the welding distortion, the u_z (Figure 2C) displacements of the plate have been

measured at some locations of the joints, by means of a CMM.

3 | FE MODEL

The numerical simulation of a welding process involves the investigation of the thermomechanical response of

the joint. This behaviour can be simulated by a numerical method, by using an uncoupled approach consisting of two consecutive analyses: the former, where the thermal problem is solved independently on the joint mechanical response, under a free-free configuration, to obtain the temperatures distribution; the latter, consisting of a subsequent mechanical analysis, where the temperatures history previously predicted at each node is used as thermal load. Such uncoupled approach, which is well established in literature for such type of analyses,^{17,22,36} allows saving computational costs with respect to the coupled one, with a comparable and an acceptable level of accuracy. All simulations have been carried out by means of the FE commercial code ABAQUS® v. 6.14.

The same FE model has been used for both thermal and mechanical analyses. Concerning the mesh, eight-node hexahedral 3D FEs have been used for both base and weld zones. More in detail, DC3D8 FEs have been used for the thermal analysis, allowing introducing the temperature as unique degree of freedom, and C3D8 FEs, characterized by the three translations as degrees of freedom, have been used for the mechanical analysis. According to Figure 3, a finer mesh has been developed for the chamfer region; a transition mesh for the HAZ region and a coarser mesh, with a linear bias, for the other parts of the plate. As a result, FE model counts a total of 11,904 FEs and 14,175 nodes.

Moreover, the simulation of the weld passes has been carried out according to the element birth and death technique. Such technique starts by the modelling of the whole weld bead (Figure 4A), by separating it in 124 groups of FEs (62 for each weld pass), named in the following ‘components’ (Figure 4B). Subsequently, all elements are deactivated by multiplying their properties by a severe reduction factor (e.g., 10^{-6}) in a way to exclude them from the simulation. When the added material needs to be simulated in order to virtually perform the welding process, it is not actually added to the model, but it is progressively reactivated, component by component: material properties of the FEs belonging to the weld seam return to their starting values, participating again to the evolution of the joint material.

The model has been developed by taking advantage of the symmetry of the joint geometry and of the use of the same material for all joint parts (weld bead included). Under this condition, it is possible to simulate the experimental test just by modelling one plate and a half seam.

Both thermal and mechanical analyses involve a first step, $1 \cdot 10^{-4}$ s long, during which all ‘components’ simulating the whole weld seam are removed. Then, 62 couples of load steps, corresponding to about 158.97 s, are alternatively set to simulate the first weld pass (Figure 4B, red elements). Each couple consists of a first step, $1 \cdot 10^{-4}$ s long, which allows reactivating a single

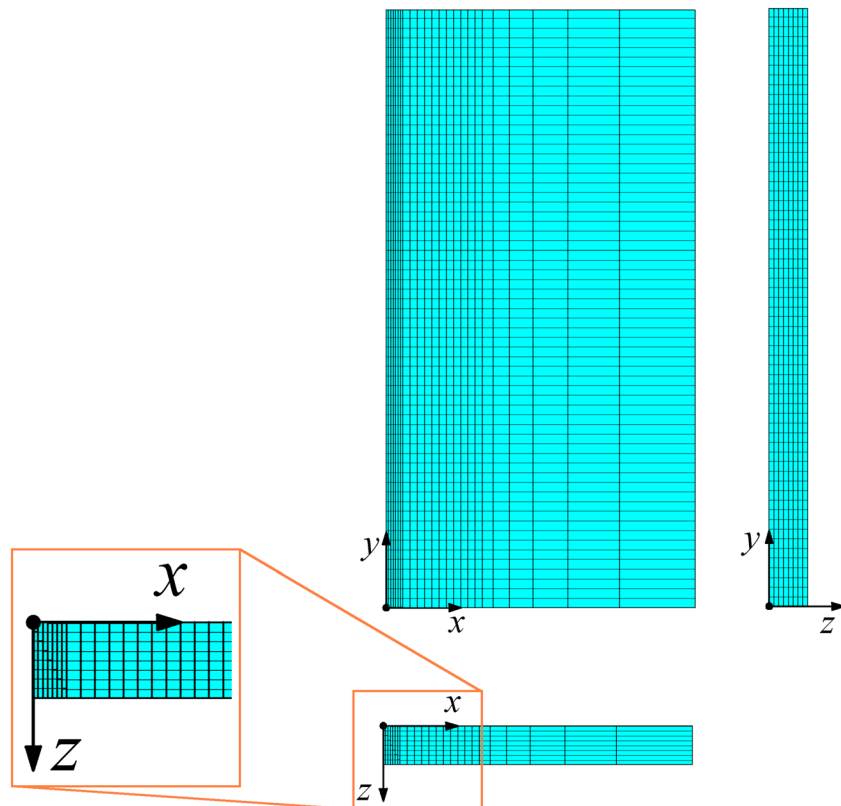


FIGURE 3 Finite element model

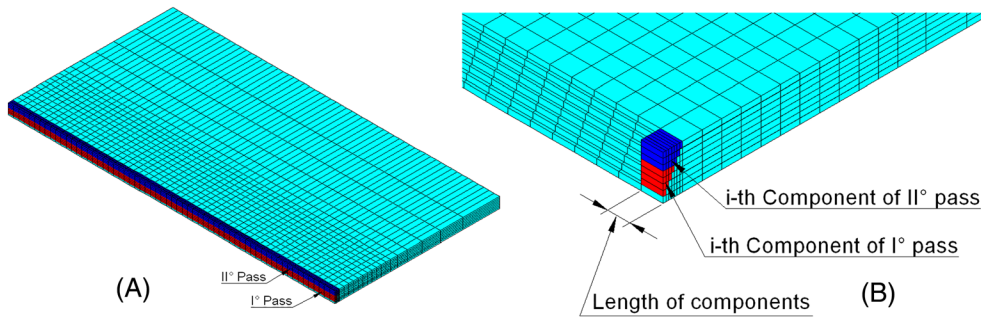


FIGURE 4 Representation of the ‘components’ with indication of the two groups of elements constituting the two passes: (A) *i*th red components of first pass and (B) *i*th blue components of second pass

‘component’ and a second step, which allows simulating the thermal load on the previously reactivated ‘component’ (the duration time, in this case, depends on the weld speed). Before simulating the second pass, a load step, 108 s long, is set to simulate the time dedicated to the first weld seam cleaning operations. Afterwards, other two groups of 62 load steps, corresponding to about 114.5 s are set to simulate the second pass (Figure 4B, blue elements) according to the same modelling technique of the first pass. Finally, a load step, 1886 s long, has been set to simulate the plate cooling phase up to a temperature of about 60°C. The FE analysis time increment is automatically calculated by the software, and a full Newton–Raphson method is used to obtain the incremental calculation.

3.1 | Thermal analysis

The thermal analysis of the welding process is essentially a mathematical solution of the differential problem based on the equation of energy conservation:

$$\rho C \frac{\partial T}{\partial t} = \dot{u}''' + \frac{\partial}{\partial x} \left(k_x \frac{\partial T}{\partial x} \right) + \frac{\partial}{\partial y} \left(k_y \frac{\partial T}{\partial y} \right) + \frac{\partial}{\partial z} \left(k_z \frac{\partial T}{\partial z} \right), \quad (2)$$

where the temperatures distribution $T(x, y, z, t)$ of the welded plate is a function of both spatial and time coordinates; ρ , C and k are the density, the specific heat and the thermal conductivity of the material, respectively; and \dot{u}''' is the change rate of internal energy per volume unit. Equation 2 is a nonlinear differential equation because ρ , C and k depend on the temperature. Initial and boundary conditions of the problem are, respectively,

$$T(x, y, z; t = 0) = T_0, \quad (3)$$

$$\dot{q}_n(x, y, z; t) = - \left(k_x \frac{\partial T}{\partial x} n_x + k_y \frac{\partial T}{\partial y} n_y + k_z \frac{\partial T}{\partial z} n_z \right), \quad (4)$$

where $T_0 = 23.5^\circ\text{C}$ is the initial temperature of the material and \dot{q}_n is the heat flux at a generic boundary having an outward local unit vector $\hat{n}(x, y, z)$. In welding problems, at external surfaces, the heat flux \dot{q}_n may consist of one or more of the following modes: convective heat loss, radiative heat loss and boundary heat \dot{q}_0 . The latter has been neglected in the proposed FE model. Convective and radiative heat losses on the external surfaces of the welded plates are given respectively by

$$\dot{q}_{nc} = h_c [T(x, y, z; t) - T_\infty], \quad (5)$$

$$\dot{q}_{nr} = \varepsilon \sigma \{ [T(x, y, z; t) - T_{az}]^4 - (T_r - T_{az})^4 \} = h_r [T(x, y, z; t) - T_r], \quad (6)$$

where T_∞ and T_r are, respectively, the temperatures of the environment transferring heat by convection and radiation, and they are usually equal to the room temperature; ε is the surface emissivity; $\sigma = 5.67 \cdot 10^{-8} \text{ W m}^{-2} \text{ K}^{-4}$ is the Stefan–Boltzmann constant; h_c is the temperature-dependent convective film coefficient; and $T_{az} = -273.15^\circ\text{C}$ is the absolute zero of the thermal scale used for this work (Celsius degrees). From Equation 6, the radiative heat losses can be expressed in the form of convective heat losses by means of temperature-dependent convective film coefficient h_r ; therefore, from Equations 5 and 6, a unique temperature-dependent film coefficient, H , can be considered:

$$H = h_c + h_r. \quad (7)$$

Particularly important in the thermal model is the heat input per millimetre Q , reported in Table 3. This is the energy supplied by the welding machine per unit of length. In the proposed simulation, a half of this energy has been supplied to a half of the seam because one plate only has been modelled. Therefore, energy supplied to the entire half welding seam during the simulation is equal to

$$Q_{\text{real}} = \frac{Q \cdot L_{\text{seam}}}{2}, \quad (8)$$

where L_{seam} is the length of welding bead.

This energy can be subdivided into three parts:

1. *Sensible heat*: energy to heat the weld material from the initial temperature (T_0) to the solidus temperature (T_s):

$$Q_{\text{sensible}} = \text{vol}_{\text{seam}} \cdot \int_{T_0}^{T_s} \rho C dT = m_{\text{seam}} \int_{T_0}^{T_s} C dT, \quad (9)$$

where ρ and C are the density and specific heat of the material, respectively, and vol_{seam} and m_{seam} are the volume and the mass of half welding bead.

2. *Latent heat*: energy due to phase transition from the solidus temperature (T_s) to the liquidus temperature (T_L):

$$Q_{\text{latent}} = m_{\text{seam}} \cdot q_{\text{latent}}, \quad (10)$$

where m_{seam} is the mass of half welding bead and q_{latent} is the latent heat per mass unit.

3. The energy to further heat the weld material is equal to

$$Q_{\text{body flux}} = Q_{\text{real}} - Q_{\text{sensible}} - Q_{\text{latent}}, \quad (11)$$

whereas the energy to be applied to the single components is

$$Q_{\text{component}} = Q_{\text{body flux}} / n_{\text{component}}, \quad (12)$$

where $n_{\text{component}}$ is the number of components of whole half welding bead.

This latter part of the energy acts as volumetric generation of the internal energy \dot{u}''' (Table 3), and it is computable by Equation 13:

$$\dot{u}''' = \frac{Q_{\text{component}}}{\text{vol}_{\text{component}} \cdot t_{\text{weld}}} = \frac{Q_{\text{body flux}}}{n_{\text{component}} \cdot \text{vol}_{\text{component}} \cdot L_{\text{component}}} = \frac{Q_{\text{body flux}} \cdot \nu}{\text{vol}_{\text{seam}} \cdot L_{\text{component}}}, \quad (13)$$

where $\text{vol}_{\text{component}}$ and $L_{\text{component}}$ are the volume and the length of the single component, respectively; ν is the welding speed; and t_{weld} is the time necessary to travel a distance equals to the length of the single component by Equation 14:

$$t_{\text{weld}} = L_{\text{component}} / \nu. \quad (14)$$

In the proposed FE model, the specific power \dot{u}''' has been applied to each component during the time t_{weld} as volumetric flux, and it has been applied by means of the law shown in Figure 5.

The load has been applied so that the area under the load curve is constant and equals to \dot{u}''' at varying travel time (t_{weld}). Two ramps (with duration of 0.5% of t_{weld}), to avoid the discontinuity during the load application, and a little time offset of $2 \cdot 10^{-6}$ s between two load curves have been defined, in order to encourage the convergence of the solution. The height of the trapezium h is computable by Equation 15:

$$\dot{u}''' = A_{\text{trapezium}} = \frac{(B+b) \cdot h}{2} = \frac{\{(1 \cdot t_{\text{weld}}) + [(1 - 2 \cdot 0.005) t_{\text{weld}}]\} \cdot h}{2} = (1 - 0.005) \cdot t_{\text{weld}} \cdot h, \quad (15)$$

and hence,

$$h = \dot{u}''' / (1 - 0.005) \cdot t_{\text{weld}}. \quad (16)$$

Thermal boundary conditions, given by Equations 3 and 4, occur at all the surfaces except the adiabatic plane

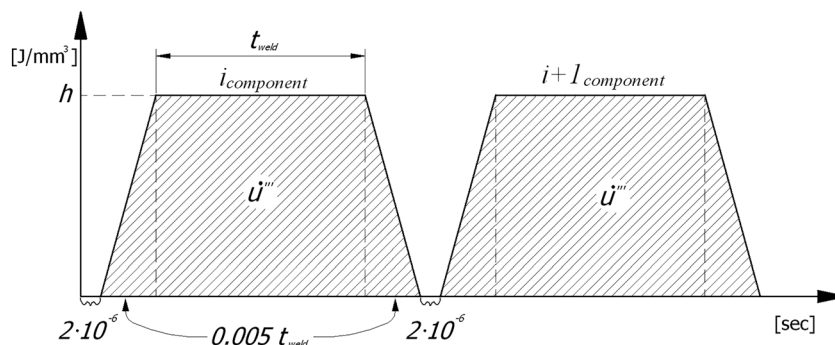


FIGURE 5 Load curve

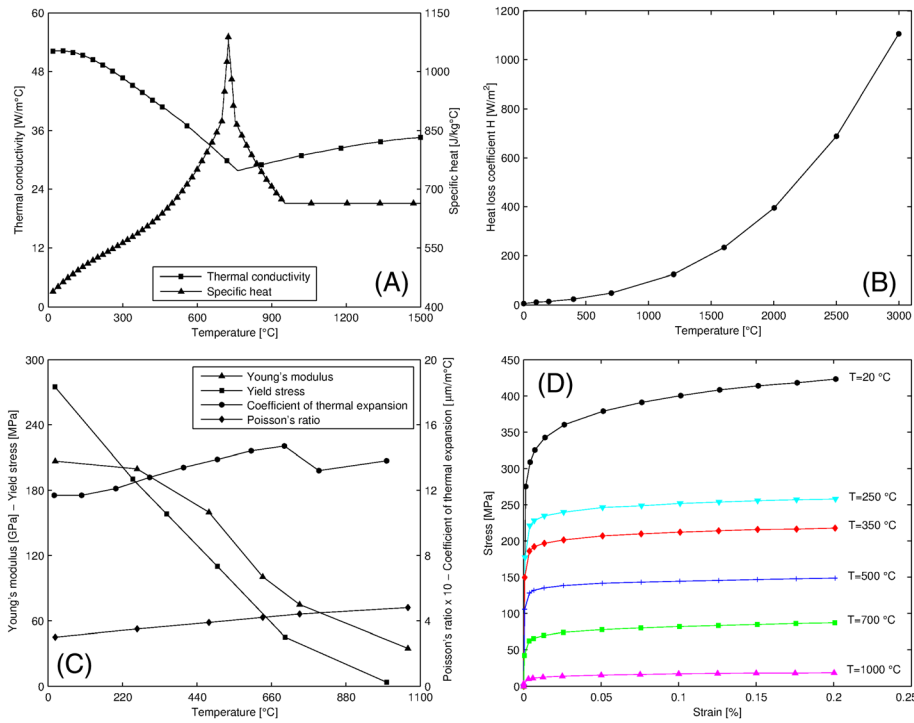


FIGURE 6 (A) Thermal properties of material dependent on temperature, (B) variation of heat loss coefficient H with temperature, (C) mechanical properties of material dependent on temperature and (D) stress versus strain curves with temperature

at the longitudinal symmetry plane of the final joint; in the same manner, for numerical simplicity, the surfaces of the chamfer hollow have been considered adiabatic before supplying the filler material.

Thermal properties dependent on temperature^{46,47} are reported in Figure 6A. In order to take into account the transformation phase in the thermal analysis, a latent heat per mass unit, a solidus temperature and a liquidus temperature equals to 277,000 J kg⁻¹, 1495°C and 1540°C, respectively, have been used, whereas the temperature-dependent heat loss coefficient H , applied on the surfaces, is shown in Figure 6B.

3.2 | Mechanical analysis

Temperature history achieved by the thermal analysis has been used in the mechanical analysis as thermal loads. An elastoplastic material model, based on the von Mises yield criterion and isotropic strain hardening rule, has been considered, including the effects of the temperature on the material properties.

The stress–strain relations can be written as

$$[d\sigma] = [D^{ep}] \cdot [d\varepsilon] - [C^{th}] \cdot dT, \quad (17)$$

being

$$[D^{ep}] = [D^e] + [D^p], \quad (18)$$

where $[D^{ep}]$ is the total stiffness matrix, $[D^e]$ is the elastic stiffness matrix, $[D^p]$ is the plastic stiffness matrix and $[C^{th}]$ is the thermal stiffness.

Moreover, as aforementioned, a particular attention must be paid on the modelling of the symmetry boundary conditions. The transient thermal field generated by the welding process introduces several deformations inside the plates, leading to their interaction. Such interaction, which increases as the plate length increases (due to their rotation), has never been considered in the FE models proposed in literature,^{17,21,22,31,36,38} by reducing this problem to a simple application of the symmetric boundary conditions. This results into too many approximations in the simulated residual stress–strain state, especially for long plates.

More in detail, when the first component is reactivated together with its symmetric boundary conditions, the plate starts to rotate due to the thermal loads, approaching, as a consequence, to the longitudinal symmetry plane. By progressively reactivating the components, up to the last components of the weld seam, the plate rotation may induce components to find themselves significantly beyond the longitudinal symmetry plane. Actually, such rotation is limited by the interaction of the plate with its counterpart. All these considerations suggest considering such phenomenon during the modelling, even if the modelling involves both plates.

Moreover, the plate rotation may lead also to convergence issues, especially as the plate length increases. The main reason of these convergence issues can be addressed to the activation of the symmetric constraints, which, under these conditions, would be applied to a more deformed plate (with respect to the real test case), leading to an increase of the residual stresses that can facilitate the lack of the analysis convergence.

In order to take into account the interaction of both plates in the proposed symmetric approach-based FE model, a row of FEs (green FEs in Figure 7) has been placed along the left side of the longitudinal symmetry plane (Figure 7A). This row of elements, of the same length as the plate (248 mm), is made of 62 C3D8 FEs and 252 nodes; an arbitrary width (x direction) and a height (z direction), slightly greater than the 'root face' (Figure 1), simulates the interaction with the two plates, also in case of out-of-plane displacements. The mechanical material properties of these FEs are the same of the plate. Concerning the boundary conditions applied on this row of elements, nodes placed on the interacting surface (face looking at the longitudinal symmetry plane) have been fully constrained.

In addition, the interaction between the plate and the work table, shown in Figure 1D, has been numerically replicated by modelling the work table as a rigid plane and by modelling the interaction between the plate and the rigid plane by means of a surface to surface contact algorithm. Moreover, in order to completely constrain the rigid motion of the plate, the translation along the y and z axes of the node ($x = 0, y = 0, z = 0$) of the seam and the translation along the z axis of the node ($x = 0, y = 248, z = 0$) have been fixed.

The interaction between the modelled plate and the row of elements has been defined through a surface to surface contact algorithm. Specifically, at the first load step the interaction involves all V-grove FEs; subsequently, due to the progressive reactivation of 'components' of the first weld pass together with the activation of the symmetric constraints along the x direction

(Figure 7), the interaction and the green FEs belonging to the row of elements are removed progressively as well (Figure 7B), because not more useful. It must be highlighted that the simulation strategy does not increase significantly the computational costs.

The translational constraints along the x direction are progressively applied to the components nodes placed along the longitudinal symmetric plane during the weld pass. This type of boundary condition constrains also the rigid motion of the plate around y and z directions.

Also, the mechanical properties^{48,49} have been considered temperature dependent (Figure 6C). Moreover, because the structural analysis involves plastic deformation of the material, in this work, the hardening material model with von Mises yield criterion and the isotropic strain hardening rule have been assumed. The stress versus strain curves, as a function of the temperature, are shown in Figure 6D.

For some steel welded parts the solid-state austenite–martensite transformation during cooling has a relevant influence on the residual stresses and distortion,^{17,33,36,50–53} especially when the equivalent carbon content is high. Instead, for low equivalent carbon content, some authors showed that the transformation phase may be neglected.^{54,55} So according to Cho and Kim⁵⁴ and Deng,⁵⁵ the modelling of the transformation phase does not change the level of accuracy of the FE model, in terms of residual stresses and distortions prediction, when the welding process involves low carbon steels with low equivalent carbon content (about $C_{eq} = 0.23\%$), as the steel used in this work. On the contrary, other authors showed the influence of such phenomenon on the residual stresses prediction also for low carbon content (Romanin et al⁵⁶ and Ferro et al⁵⁷). However, in this paper, the solid-state transformation phase has not been considered in the FEM modelling.

In order to evaluate the influence of the interaction between the plates, a second mechanical analysis has been carried out without considering the row of FEs

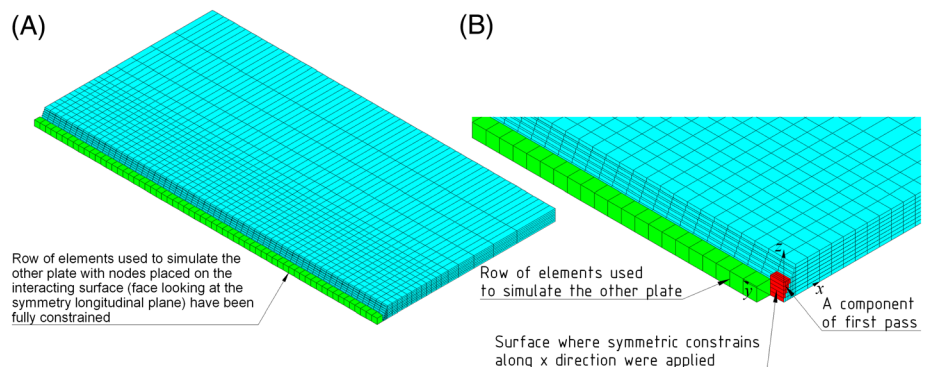


FIGURE 7 (A) Row of elements (in green) simulating the left side plate and (B) suppression of the elements of row activation of the symmetric constraints along x direction

placed along the left side of the longitudinal symmetry plane (Figure 7) and by applying only the translational constraints along the x direction to simulate the symmetry constraints.

4 | RESULTS AND DISCUSSION

In this section, numerical results of the welding simulation are presented and compared with experimental ones, in order to assess the reliability of the used FE models.

4.1 | Thermal analysis

The temperatures measured at six points by means of six thermocouples have been compared with the respective predicted ones.

Figure 8 shows the temperature distributions at the middle section of the plate along the transverse direction during the two passes, with the welding arc located at the middle of the welding path. It is possible to appreciate a good agreement between numerical and experimental results, with the numerical curves that well estimate the experimental measurements during both welding passes.

The experimental-numerical thermal histories are shown in Figure 9 for the thermal cycles recorded in correspondence of the thermocouples TC1 and TC2 and in Figure 10 for thermal cycles recorded from thermocouples from TC3 to TC6.

Figure 9 shows a good agreement between numerical and experimental results as general trend. In order to better highlight the differences between numerical and experimental curves, thermal histories curves at TC1 and TC2 have been cut at a time of 550 s and reported in Figure 9B,D, respectively. According to Figure 9A,B, the

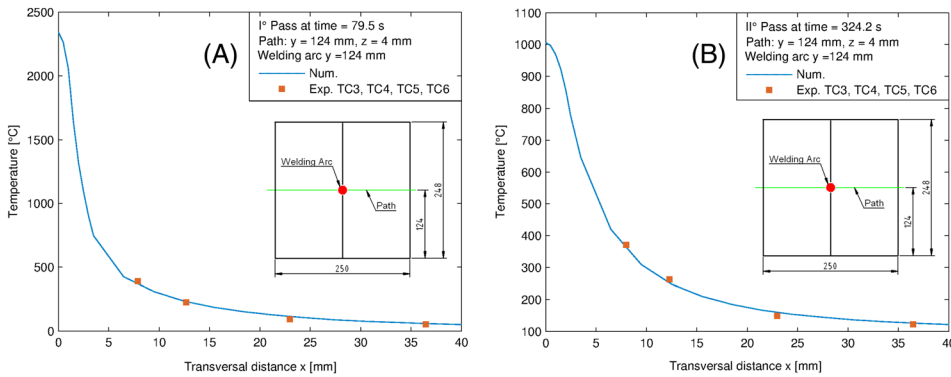


FIGURE 8 Temperature distributions at the middle section of the plate along transversal direction during (A) the first and (B) the second passes

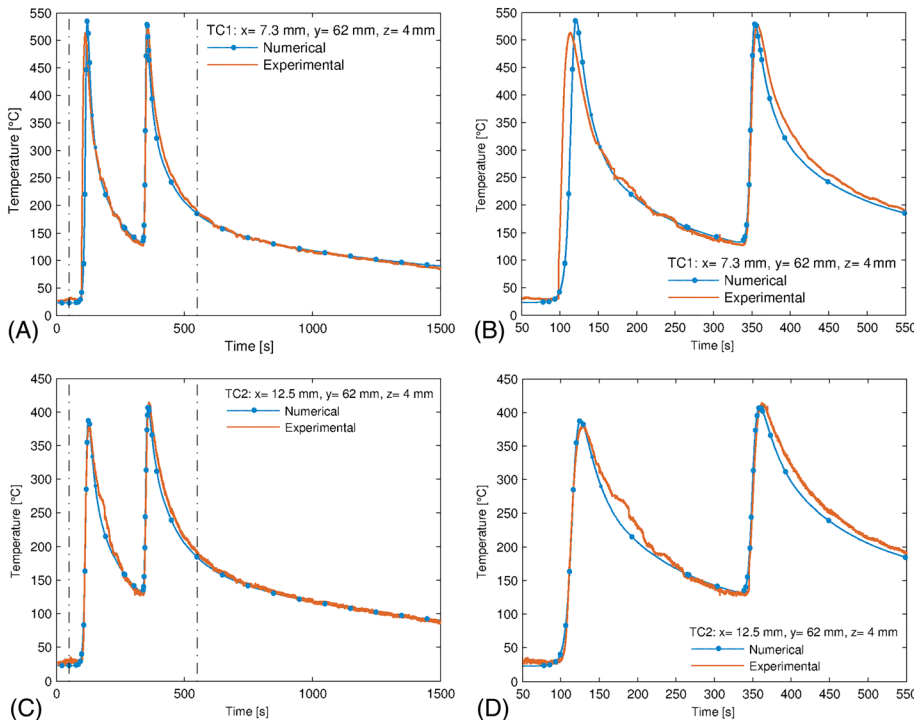
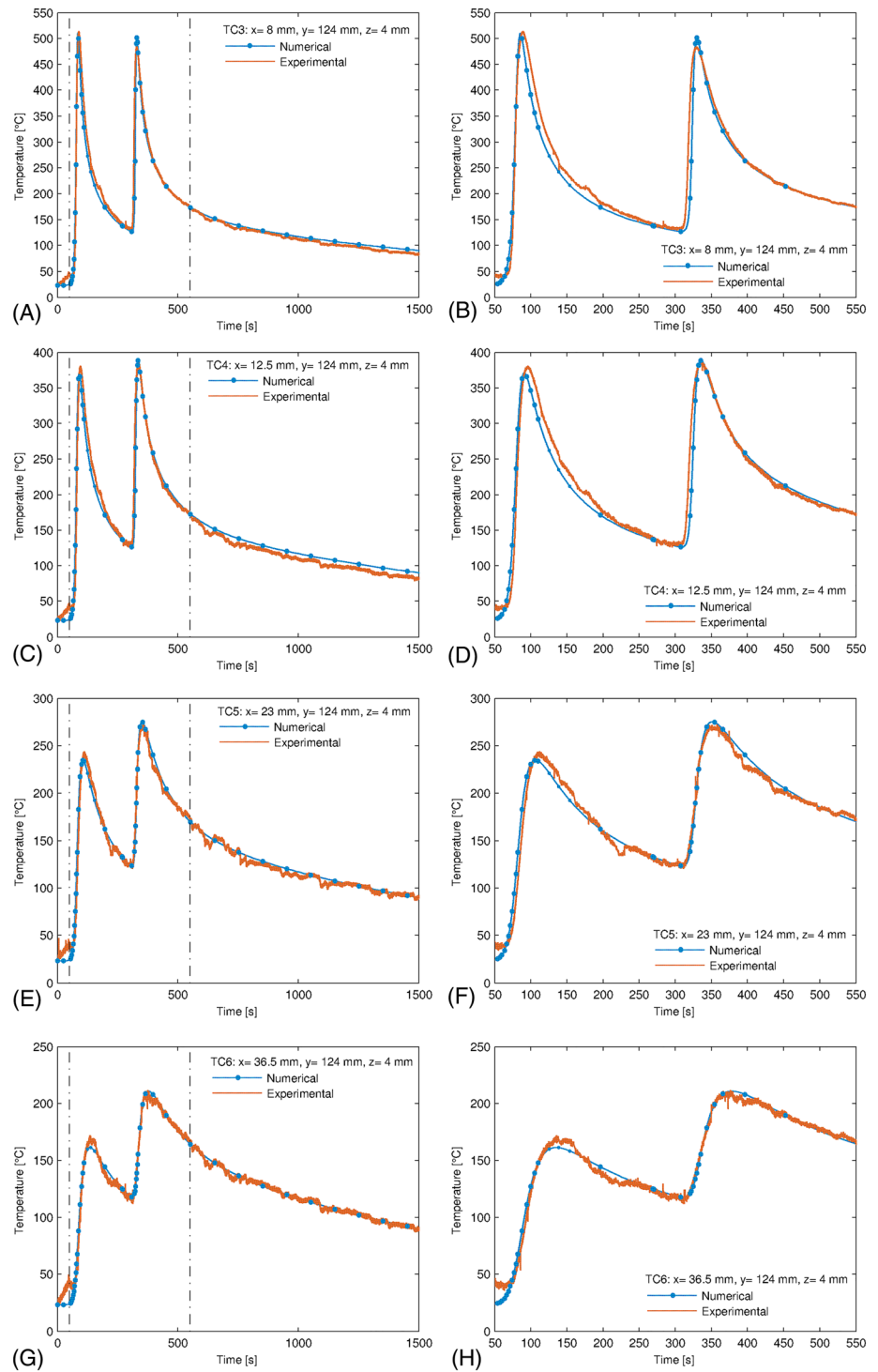


FIGURE 9 Thermal histories in correspondence of thermocouples: (A) TC1, (B) focus on TC1, (C) TC2 and (D) focus on TC2

FIGURE 10 Thermal histories in correspondence of thermocouples: (A) TC3, (B) focus on TC3, (C) TC4, (D) focus on TC4, (E) TC5, (F) focus on TC5, (G) TC6 and (H) focus on TC6



FE model provides, in correspondence of TC1, a small overestimation of about 23°C in the first peak value, corresponding to the first pass. This can be due to a displacement of the thermocouple within the plate hole, occurred during the first pass, leading to a not perfect contact between the thermocouple and the plate. This difference cannot be found during the second pass because the correct position of all thermocouples was checked at the end of the first pass.

Moreover, according to Figure 9B, a time shift can be observed between numerical and experimental data. Such disagreement can be attributed to the welding speed, not perfectly constant along the whole weld seam, but, in proximity of thermocouple TC1, faster than the value reported in Table 3. As matter of the fact, the welding speed reported in Table 3 has been calculated as the ratio of the time spent for the first pass to the length of the weld seam.

Figure 10 shows the comparison of the numerical results with the experimental ones in correspondence of thermocouples located at the middle section of the plate during the whole process. Also here, in order to highlight the differences between numerical and experimental results, curves have been cut at 550 s and reported in Figure 10B,D,F,H. Excluding the initial experimental values of the curves, which are affected by high noise due to the electrical shock at the beginning of the welding process, it can be stated that the experimental results are in good agreement with the numerical ones. It can be observed that the FE model provides a small overestimation of about 17°C in correspondence of the second peak, during the second pass, at TC3 (Figure 10B), whereas there are slight differences in the peak values during the first pass at TC4 (Figure 10D) and TC6 (Figure 10H), characterized by an underestimation of the numerical results. However, these differences are lower than about 5%.

Figure 11A,B shows the temperature distributions during the first and second passes at welding times of $t = 66.67$ s and $t = 315$ s, respectively. The centre of the welding arc at these times is at the position of $x = 0$ mm and $y = 104$ mm. As it can be seen from the figures, the peaks of temperature around the welding arc, at the two instants, are calculated to be about 2460°C and 2116°C , respectively, suggesting that the material is melted in the

FZ. High temperatures are present nearby the FZ, defining the HAZ.

During the first pass, the thermal gradients are very steep in the proximity of the heat source, whereas they decrease during the second pass; furthermore, near the welding line, the distance between the isotherms increases as the cooling rate decreases and the welding pool also presents a very small area in front of the welding arc.

Figure 12 compares the experimental and the predicted FZ and HAZ. As shown in this figure, the geometry and shape of both weld seam and HAZ are well numerically replicated. As matter of the fact, the HAZ can be defined as the section reaching a temperature higher than 727°C during the welding process.⁵⁸

By comparing the numerical results with the experimental ones, it is possible to state that the used methodology is suitable to predict very accurately the temperatures distribution in the welded joint, without the need to perform a tuning process to opportunely calibrate the heat source.^{17,27,32,35,36,46}

4.2 | Mechanical analysis

The longitudinal residual stresses σ_y , induced from the longitudinal expansion and contraction of the material

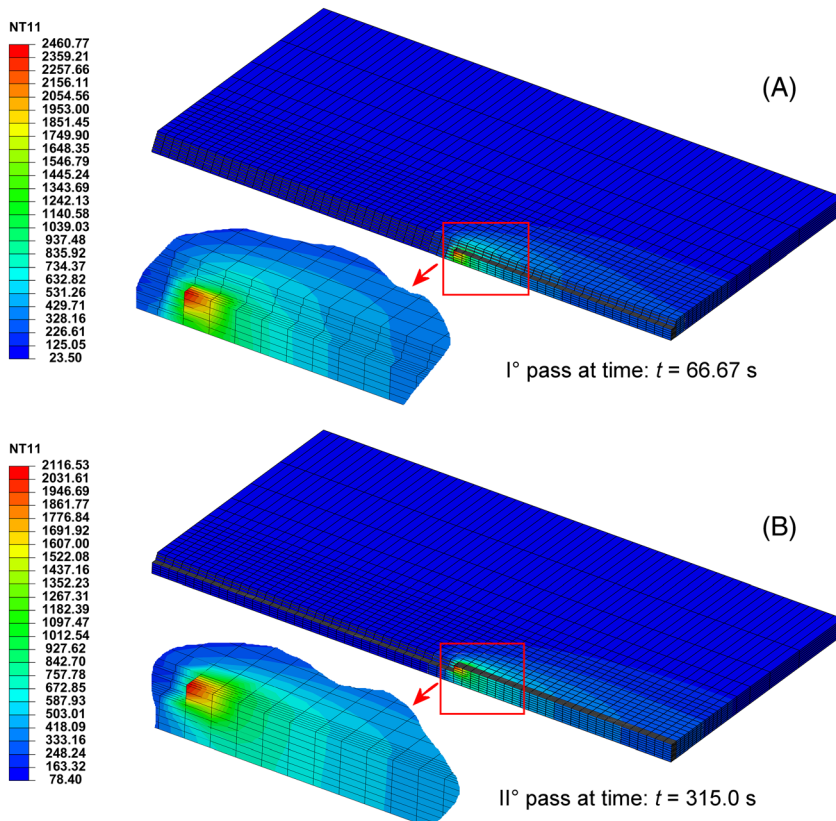


FIGURE 11 Temperature distributions ($^{\circ}\text{C}$) during (A) I pass at time $t = 66.67$ s and (B) II pass at time $t = 315$ s

FIGURE 12 Numerical-experimental comparison of heat-affected zone (HAZ), during (A) the first and (B) the second pass

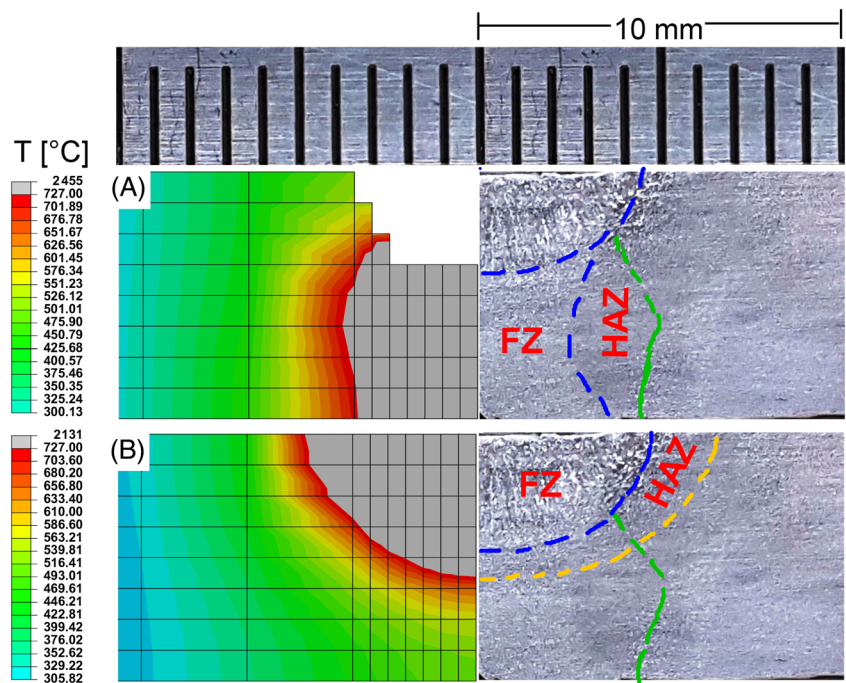
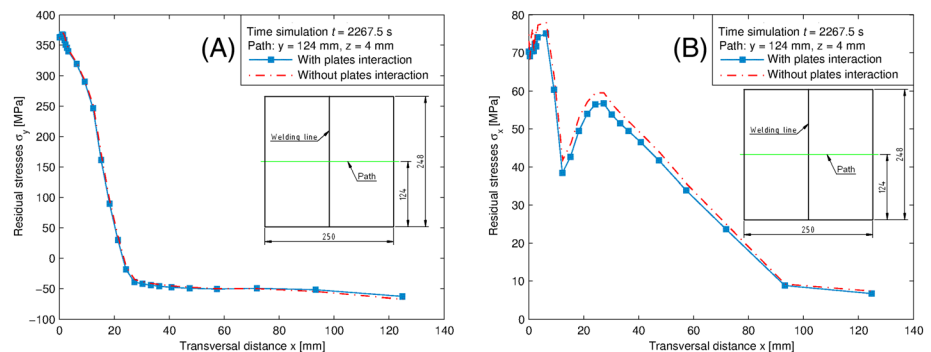


FIGURE 13 Comparison of (A) the longitudinal residual stresses distribution σ_y and (B) the transversal residual stresses distribution σ_x , along x direction at midsection ($y = 250$ mm, $z = 4$ mm) achieved by the finite element (FE) models considering and not the plates interaction



during the welding process, along the x direction at the midsection ($y = 124$ mm, $z = 4$ mm) are shown in Figure 13A (line with marker). The self-equilibrium of the weldment is such that the tensile and compressive residual stresses are present at the weld seam and away from the welding line respectively. High tensile residual stresses are present in correspondence of the zones nearby the welding centre line (WCL), due to the contraction resistance of the material as the cooling phase begins. Then, they decreased to zero, as the distance from the (WCL) increases, becoming compressive for the zones far from the weld seam. The transversal residual stresses σ_x along the x direction in the midsection ($y = 124$ mm, $z = 4$ mm) are shown in Figure 13B. Tensile residual stresses are present in correspondence of the zones nearby the (WCL). Subsequently, they decreased as the distance from the (WCL) increased, up to 0, almost.

Figure 14 shows the residual stresses map at the end of cooling time. As it is possible to observe, the zones

close to the welding line are subjected to high values of tensile longitudinal residual stresses σ_y , whereas the zones far from welding line are subjected to compressive residual stresses (Figure 14A). In particular, the values of σ_y in almost the whole welding seam are higher than the yield stress of material. Concerning the transversal residual stresses σ_x , Figure 14B shows that the central part of weld seam is subjected to tensile stresses whereas the zones closer to the ends of the welding seam are subjected to compressive stresses. Finally, it can be observed that the residual stresses distribution are about symmetrically with respect to the middle plane at $y = 124$ mm.

Figure 15 shows the out-of-plane deformation u_z contour map at the end of cooling time, due to a non-uniform thermal contraction through plate thickness.

In Figure 16, the numerical displacements, u_z , predicted in correspondence of the path $y = 248$ mm, $z = 0$ mm (green line of Figure 16A) and in corresponding of the path $x = 125$ mm, $z = 0$ mm (red line of Figure 16B), achieved

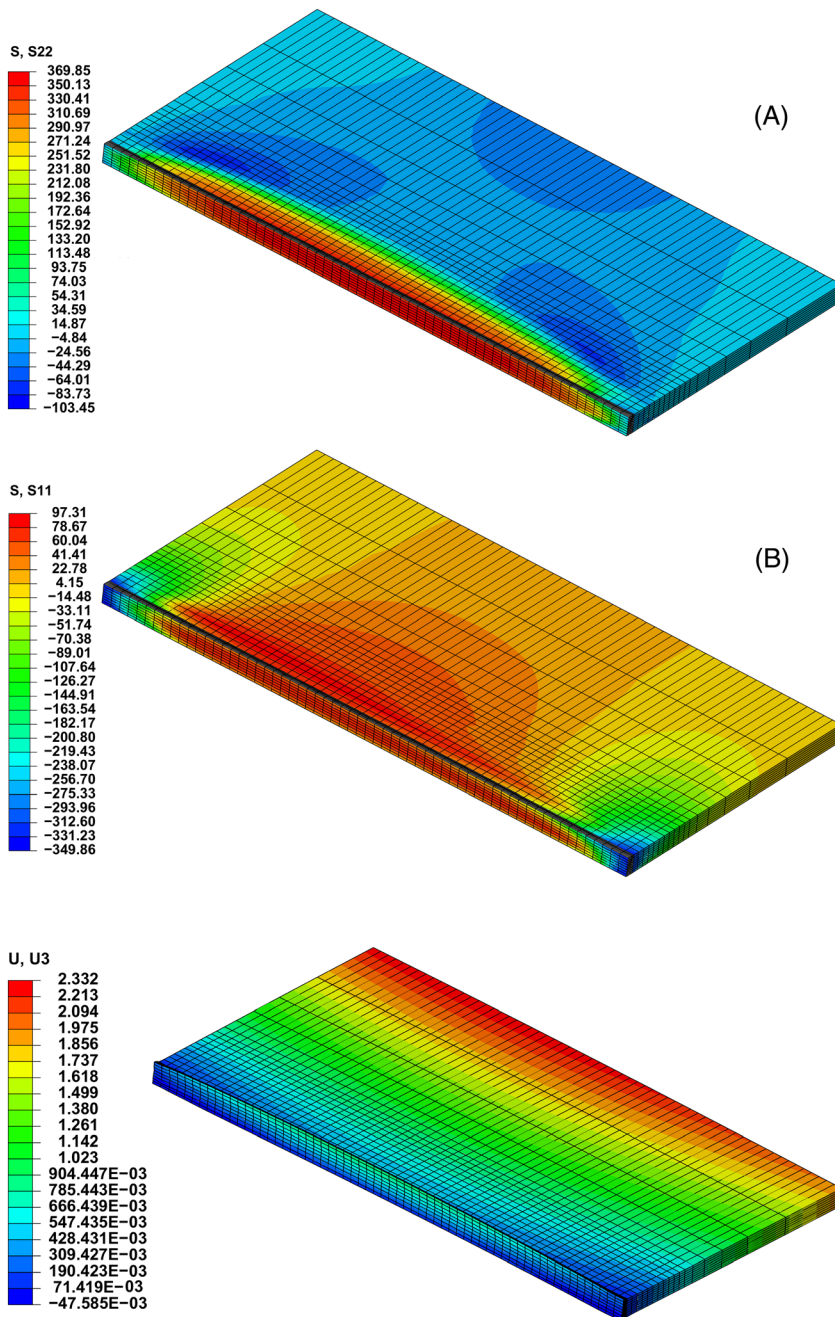


FIGURE 14 Residual stresses distributions (MPa): (A) longitudinal residual stresses σ_y and (B) transversal residual stresses σ_x

FIGURE 15 Displacements u_z (mm) at the end of cooling time

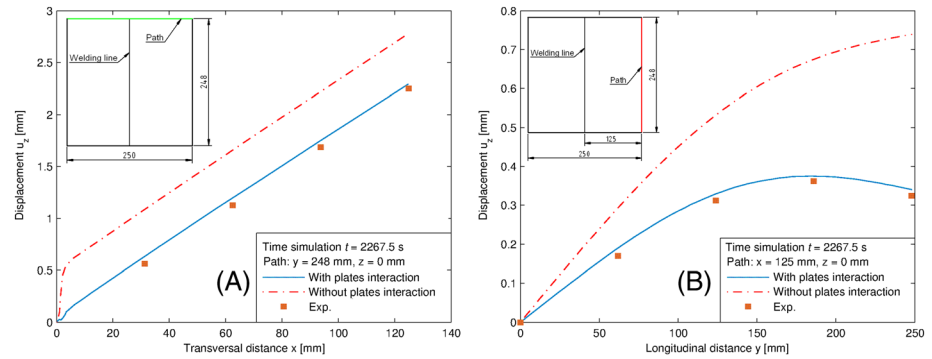
by imposing the temperatures distribution predicted by the thermal analysis as thermal loads in the mechanical analysis, are compared with the experimental ones measured by means of a CMM shown in Figure 2C. According to Figure 16, the numerical results are in good agreement with experimental ones; therefore, it is possible to state that the numerical technique, used for the mechanical analysis, can predict with a high level of accuracy the distortions in the welded joint.

As aforementioned, in order to appreciate the effects provided by the modelling of the plates interaction during the welding process, the predicted residual stresses and distortions have been compared with those provided

by the simulation performed by deactivating the plates interaction (Figures 13 and 16).

According to Figure 13A, as expected, it can be noticed that the longitudinal residual stresses distribution seems to be unaffected by the plates interaction. As matter of the fact, plates have not been constrained along the longitudinal direction. Contrary, the plates interaction affects the transversal residual stresses distributions, because of the plates rotation during the welding process. Residual stresses appear to be slightly higher (Figure 13B) for the model that does not consider the plates interaction, and they are expected to increase for longer plates, because of their rotation.

FIGURE 16 Comparison of the experimental distortions with those predicted by the finite element (FE) models considering and not the plates interaction: (A) along the path at $y = 248$ mm and $z = 0$ mm and (B) along the path at $x = 125$ mm and $z = 0$ mm



If the effects of the plates interaction on residual stresses distribution may be considered negligible for the selected test case, a similar consideration cannot be done in terms of distortions distribution. According to Figure 16, the predicted distortions distribution appears to be sensibly higher and far from the experimental data for the FE model that does not consider the plates interaction. As a result, the plates interaction plays a key role in the modelling of the welding process induced distortion.

5 | CONCLUSIONS

This paper presents a novel numerical model, based on the FEM, for the simulation of a welding process aimed to make a two-pass V-groove butt weld joint. In order to evaluate the residual stresses, a 3D nonlinear thermomechanical analysis has been carried out. The thermomechanical response of the joint has been simulated by using an uncoupled approach. Specifically, the ‘element birth and death’ technique has been used to simulate the welding filler during the welding process. The originality of the proposed technique has to be found in the simulation of the interaction occurring between the two plates during the welding process, never considered in literature when the problem is faced through a symmetrical approach. As a result, it was possible to predict more accurately the residual stresses affecting the joint, caused by the thermal distortions which lead the plates to rotate. The proposed modelling technique appears to be fundamental for long plates, because the plates interaction becomes not negligible as the plate length increases. Specifically, in order to save the computational costs, only a plate and half seam have been modelled. As a result, in order to simulate the plates interaction, a row of FEs has been placed along the left side of the longitudinal symmetry plane. This approach allows predicting the residual stresses also for long joined plates, which require a higher number of nodes and elements and, consequently, a higher time analysis. A

surface to surface contact algorithm has been considered between the half seam and the FEs row.

Moreover, differently from the literature, the heat amount is supplied to the FEs as a volumetric generation of the internal energy, allowing overcoming the time-consuming calibration phase required by the Goldak’s model, commonly adopted in literature.

The reliability of the FE model has been shown by assessing the predicted results, in terms of temperatures distribution and joint distortion, against the results provided by an experimental test. Temperatures distribution has been measured during the welding process by using six thermocouples placed at different locations nearby the weld bead; welding distortions were measured by means of a CMM. A good agreement has been found between numerical and experimental results, showing the effectiveness of the proposed FE modelling technique.

NOMENCLATURE

C	specific heat
C_{eq}	equivalent carbon content
CMM	coordinate measuring machine
$[C^{th}]$	thermal stiffness
DBEM	dual boundary elements method
$[D^e]$	elastic stiffness matrix
$[D^{ep}]$	total stiffness matrix
$[D^p]$	plastic stiffness matrix
E	Young modulus
FEM	finite element method
FZ	fusion xone
G	tangential modulus
GMAW	gas metal arc welding
H	temperature-dependent film coefficient
HAZ	heat-affected zone
h_c	temperature-dependent convective film coefficient
I	welding current
k	thermal conductivity
$L_{component}$	length of the single component
L_{seam}	length of welding bead

MIG	metal inert gas
m_{seam}	mass of half welding bead
$n_{\text{component}}$	number of components of whole half welding bead
q_{latent}	latent heat per mass unit
Q	heat input
$Q_{\text{component}}$	energy to be applied to the single components
Q_{latent}	latent heat
Q_{real}	energy supplied to the entire half welding seam
Q_{sensible}	sensible heat
\dot{q}_n	heat flux
SMAW	shielded metal arc welding
t_{weld}	time necessary to travel a distance equals to the length of the single component
$T(x, y, z, t)$	temperatures distribution of the welded plate
T_0	initial temperature
T_{∞}	temperature of the environment transferring heat by convection
T_{az}	absolute zero of the thermal scale used for this work (Celsius degrees)
T_r	temperature of the environment transferring by radiation
T_s	solidus temperature
\dot{u}'''	volumetric generation of the internal energy
v	the welding speed
$vol_{\text{component}}$	volume of the single component
vol_{seam}	volume of half welding bead
V	voltage
ε	surface emissivity
η	arc efficiency
ν	Poisson ratio
ρ	density
σ	Stefan–Boltzmann constant
σ_r	tensile strength
σ_s	yield stress

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How to cite this article: Sepe R, De Luca A, Greco A, Armentani E. Numerical evaluation of temperature fields and residual stresses in butt weld joints and comparison with experimental measurements. *Fatigue Fract Eng Mater Struct*. 2020;1-17. <https://doi.org/10.1111/ffe.13351>